

Water–Gas Shift Reaction on Metal Nanoclusters Encapsulated in Mesoporous Ceria Studied with Ambient-Pressure X-ray Photoelectron Spectroscopy

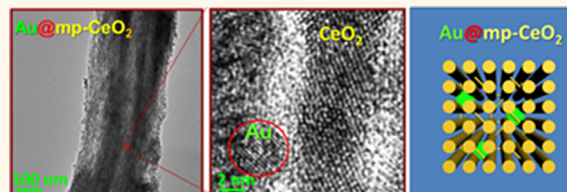
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Mesoporous materials have attracted much interest since MCM-41 was discovered in 1992.^{1,2} As a result of intense research efforts, a broad spectrum of mesoporous materials have been synthesized, ranging from silica to metal oxides.^{3–6} Currently, the affluent mesoporous materials are playing key roles in many fields, such as the fuel production and environmental protection, which fully benefit from the unique properties of the mesoporous materials.^{7,8}

One property of the mesoporous materials under intensive studies is the confinement effect, which is due to the small physical size of the pores. The spatial confinement of the nanoscale channel in mesoporous materials creates spatial selectivity between two reactants or between the reactant and active sites.^{9–11} For instance, the enantiomeric excess (ee) value in the epoxidation of non-functionalized olefins on mesoporous Mn catalysts increases with decreasing pore size because the small pores could enhance spatial recognition.¹² Recent studies showed that Fe catalysts encapsulated in carbon nanotubes (CNTs), whose pore sizes are similar to mesoporous material, have an enhanced catalytic activity in Fischer–Tropsch synthesis (FTS).^{13–15} There was some peculiar interaction between the catalyst nanoclusters and the inner wall of the CNT resulting from electron transfer driven by a difference in electron density between the external surface and internal wall of a CNT. For catalysis in CO hydrogenation, this unique interaction could promote the CO dissociation to accelerate the overall reaction rate of FTS.

ABSTRACT



Metal nanoclusters (Au, Pt, Pd, Cu) encapsulated in channels of mesoporous ceria (*mp*-CeO₂) were synthesized. The activation energies of water–gas shift (WGS) reaction performed at oxide–metal interfaces of metal nanoclusters encapsulated in *mp*-CeO₂ (*M@mp*-CeO₂) are lower than those of metal nanoclusters impregnated on ceria nanorods (*M/rod*-CeO₂). *In situ* studies using ambient-pressure XPS (AP-XPS) suggested that the surface chemistry of the internal concave surface of CeO₂ pores of *M@mp*-CeO₂ is different from that of external surfaces of CeO₂ of *M/rod*-CeO₂ under reaction conditions. AP-XPS identified the metallic state of the metal nanoclusters of these WGS catalysts (*M@mp*-CeO₂ and *M/rod*-CeO₂) under a WGS reaction condition. The lower activation energy of *M@mp*-CeO₂ in contrast to *M/rod*-CeO₂ is related to the different surface chemistry of the two types of CeO₂ under the same reaction condition.

KEYWORDS: catalysis · water–gas shift · ambient-pressure X-ray photoelectron spectroscopy · ceria · *in situ*

However, mesoporous materials still lack fundamental studies toward understanding the enhancement of catalytic performances from the point of view of the surface chemistry of the internal concave surface of pores of mesoporous materials. For example, a concave internal surface could potentially increase the repulsion of adsorbates or intermediates formed during catalysis related to the limited concave surface. This effect on a catalytic reaction, which involves the participation of the internal surface of porous

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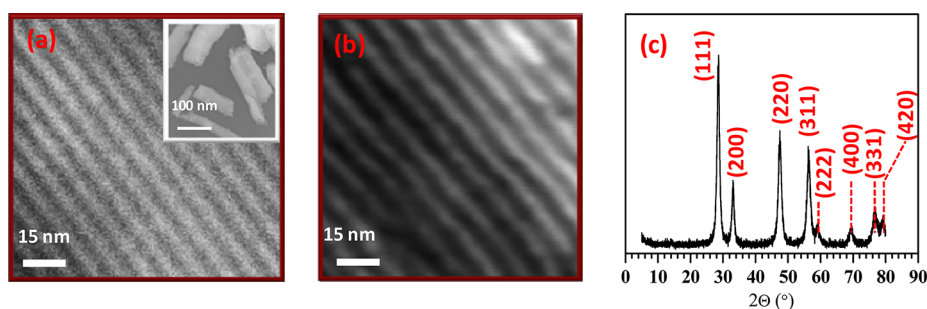


Figure 1. SEM images of SBA-15 (a) and mp -CeO₂ (b), and XRD pattern of the synthesized mp -CeO₂ (c).

particles, is expected to be greater than catalysis in which an inert support such as silica is not involved. Herein, we bring forward a fundamental study to understand the role of an *internal concave* surface of reducible oxide such as mp -CeO₂. Rod-CeO₂ with an external open surface was used for comparison.

The chosen reaction to examine the role of the internal concave surface is the water–gas shift (WGS) reaction. One reason is that CeO₂ is one of the active components of this reaction.¹⁶ It is well acknowledged that the oxygen vacancies dissociate H₂O molecules and thus provide an OH group for coupling between an OH group and an adsorbed CO molecule to form an intermediate toward dissociation and a following formation of CO₂.^{16–18} Another reason that the effort to understand the catalytic mechanism of the WGS on metal–oxide bifunctional catalysts, mostly called low-temperature WGS catalysts, is ongoing^{17,19–25} as an active, robust low-temperature WGS catalyst is critical for on-site purification of H₂ source in the large-scale applications of low-temperature fuel cell technology. Au, Pt, Pd, and Cu are chosen to study since they are active for WGS, and nanoclusters of these metals supported on oxide were well studied.

We synthesized metal nanoclusters supported on internal concave surfaces of CeO₂ pores of mp -CeO₂ and the external surface of rod-CeO₂. One of the key factors in understanding heterogeneous catalysis is the surface chemistry of the active phase during catalysis. It is well acknowledged that surface chemistry of an active phase of catalysis is determined by reaction conditions.^{26–30} Thus, an exploration of surface chemistry of catalysts under reaction conditions through *in situ* studies is significant for understanding the catalytic mechanism. Here we used ambient-pressure XPS technique^{31–33} to track the surface chemistry of catalysts during catalysis (in the presence of reactant gases at reaction temperatures).

Kinetic studies suggested lower activation energy of $M@mp$ -CeO₂ than that of M/rod -CeO₂. AP-XPS studies showed that the internal concave surface of CeO₂ pores of mp -CeO₂ of $M@mp$ -CeO₂ has a higher concentration of oxygen vacancies under a WGS reaction condition than the external CeO₂ surface of M/rod -CeO₂. A correlation between the surface chemistry of

catalysts and the corresponding catalytic properties was built. Higher concentration of oxygen vacancies of mp -CeO₂ in contrast to rod-CeO₂ under WGS reaction conditions probably results from a low adsorption energy of OH groups on the internal concave surface of pores of mp -CeO₂ due to the enhanced inter-adsorbate repulsion on the bent internal surface of mp -CeO₂.

RESULTS AND DISCUSSION

The template of mp -CeO₂, SBA-15, was synthesized and is shown in Figure 1a. Figure 1b is the SEM image of mp -CeO₂ prepared with the synthesized SBA-15. XRD patterns clearly show the crystallization of mp -CeO₂ (Figure 1c). The impregnation of metal nanoclusters in the internal wall of channels of mp -CeO₂ was performed. Pore size of mp -CeO₂ does not change upon the encapsulation of metal nanoclusters. Figure 2 presents TEM images of Au@ mp -CeO₂ and Pt@ mp -CeO₂. The lattice fringes of the metal nanoclusters were clearly identifiable and are shown in Figure 2a,2,b. For comparison, high-resolution TEM images of Au/rod-CeO₂ and Pt/rod-CeO₂ are presented in Figure 3a,b. TEM images of other catalysts are presented in Figures S5 and S6 in Supporting Information. Table 1 lists the average size of metal nanoclusters encapsulated in mp -CeO₂ and supported on rod-CeO₂. As the pores of mp -CeO₂ encapsulating different metal clusters are similar, the average sizes of Au, Pt, Pd, and Cu in mp -CeO₂ are very similar, 3–6 nm. Compared to the size of nanoclusters encapsulated in mp -CeO₂, metal nanoclusters deposited on rod-CeO₂ have a size of 4–6 nm. Thus, there is no significant difference between the size of metal nanoclusters in mp -CeO₂ and those on rod-CeO₂.

As shown in Figure 4, activation energies of WGS on mp -CeO₂ supported catalysts Pt@ mp -CeO₂, Cu@ mp -CeO₂, Pd@ mp -CeO₂, and Au@ mp -CeO₂ are lower than those on metal nanoclusters supported on surfaces of rod-CeO₂. For Pt@ mp -CeO₂, the activation energy is ~ 60 kJ mol⁻¹, lower than ~ 78 kJ mol⁻¹ for Pt/rod-CeO₂. The measured ~ 78 kJ mol⁻¹ is consistent with 74.8 kJ mol⁻¹ reported in the literature.¹⁷ A similar difference was observed on Cu@ mp -CeO₂ (24 kJ mol⁻¹) and Cu/rod-CeO₂ (39 kJ mol⁻¹) (Figure 4). The measured activation energies of WGS on Cu/rod-CeO₂ are consistent

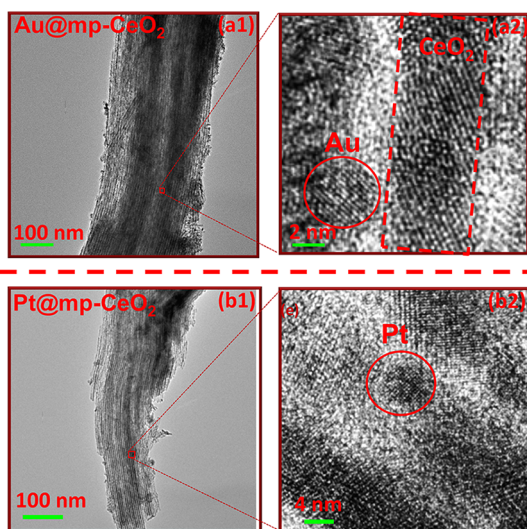


Figure 2. TEM images of the synthesized Au@*mp*-CeO₂ and Pt@*mp*-CeO₂. (a1) Large-scale image of Au@*mp*-CeO₂; (a2) high-resolution image of Au@*mp*-CeO₂ showing the encapsulation of Au nanoclusters; (b1) large-scale image of Pt@*mp*-CeO₂; (b2) high-resolution image showing the encapsulation of Pt nanoclusters.

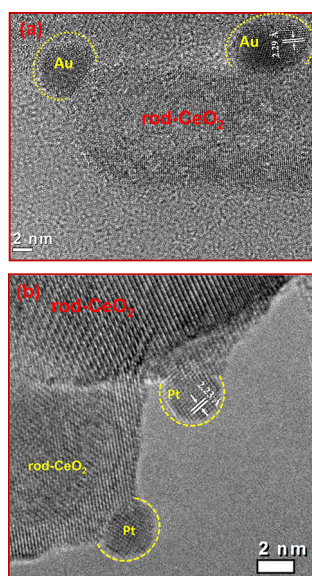


Figure 3. High-resolution TEM images of Au nanoclusters on rod-CeO₂ (a) and Pt nanoclusters on rod-CeO₂ (b).

with other reports such as 36 kJ mol⁻¹ for Cu on CeO₂ in ref 21. The low activation energies of WGS on other catalysts encapsulated in *mp*-CeO₂ including Au@*mp*-CeO₂ and Pd@*mp*-CeO₂ were revealed, as well. The activation energy of WGS on Au@*mp*-CeO₂ is 11 kJ mol⁻¹, in contrast to 51 kJ mol⁻¹ of Au/rod-CeO₂.³⁴ It is 34 kJ mol⁻¹ on Pd@*mp*-CeO₂ in contrast to 76 kJ mol⁻¹ on Pd nanoclusters supported on rod-CeO₂ and nanoparticles.³⁵

To exclude a potential mass transfer limit in the kinetic studies of WGS, kinetic studies of M@*mp*-CeO₂ by changing the flow rate of reactants were performed. No dependence of measured activation energy on

TABLE 1. Sizes of Metal Nanoclusters Encapsulated in *mp*-CeO₂ and Supported on Rod-CeO₂

metal	support	average size (nm)
Au	<i>mp</i> -CeO ₂	4 ± 1
Au	rod-CeO ₂	4.5 ± 1.5
Pt	<i>mp</i> -CeO ₂	4 ± 1
Pt	rod-CeO ₂	4 ± 1
Pd	<i>mp</i> -CeO ₂	4 ± 1.5
Pd	rod-CeO ₂	5 ± 1.5
Cu	<i>mp</i> -CeO ₂	4 ± 1
Cu	rod-CeO ₂	4.5 ± 2

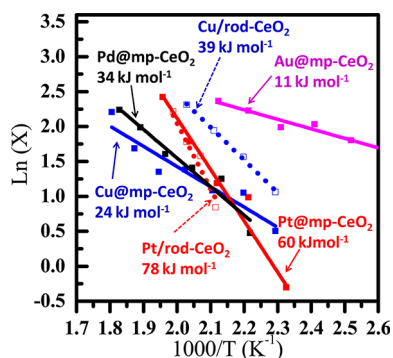


Figure 4. Arrhenius plot of WGS rates on metal nanoclusters supported on *mp*-CeO₂ and rod-CeO₂. Reaction condition and metal loading are the same for these catalysts. The reaction condition is fully described in the Methods section.

flowing rates was found. In addition, the potential influence of mass diffusion limit in the channels of *mp*-CeO₂ of M@*mp*-CeO₂ was tested by measuring the catalytic activity of catalyst particles of metal@*mp*-CeO₂ with different lengths between 100 nm and 10 μm. Catalyst activity would change with the length of the pore of *mp*-CeO₂ if mass transfer could be limited by an internal diffusion. However, our results showed that activities of these catalysts were very similar, although their particle sizes (essentially length of *mp*-CeO₂ particles) are different. This indicates that the influence of the internal diffusion on the catalyst activity is negligible. In addition, if the decrease of activation energies of M@*mp*-CeO₂ compared to those of M/rod-CeO₂ for the WGS reaction could result from a potential mass transfer limitation instead of an intrinsic property of the *mp*-CeO₂, activation energy of catalytic reactions such as CO oxidation performed on metal nanoclusters such as Pt encapsulated in an insert mesoporous material such as SBA-15 should be lower than that of Pt nanoparticles simply supported on silica particles. In fact, the measured activity energy of CO oxidation on Pt nanoclusters (2–7 nm) encapsulated in SBA-15 is similar to that of Pt on SiO₂ particles.³⁶ Thus, there is no mass transfer limitation for catalytic reaction performed in the M/*mp*-CeO₂.

The low WGS activation energies in the case of M@*mp*-CeO₂ catalysts in contrast to M/rod-CeO₂ indicate

a unique structure of the internal concave surfaces of the CeO₂ pores of *mp*-CeO₂. X-ray diffraction (XRD) patterns show that both *mp*-CeO₂ and rod-CeO₂ have face-centered cubic crystal structure (*fcc*, *Fm3m*, JCPDS 34-0394) (Figures 1c and S3a). It is expected that there is a strong metal support interaction between metal nanoclusters and CeO₂ as reported in the literature.^{18,37,38} The strong interaction between the support and metal could weaken the neighboring Ce–O bond and thus modify the mobility of oxygen atoms.³⁷ The interaction between metal atoms and the local interface of *mp*-CeO₂ is similar to the interaction between metal atoms and the local interface of rod-CeO₂. Thus, we expect the strong metal–oxide interaction to play a similar role for metal nanoclusters on *mp*-CeO₂ and rod-CeO₂. Alternatively, the confinement effect of *mp*-CeO₂ is probably the main reason for the difference in catalytic properties.

Although *mp*-CeO₂ and rod-CeO₂ have a similar crystal structure, the surfaces of *mp*-CeO₂ and rod-CeO₂ are different. The scanning transmission electron microscopy (STEM) images indicated that the *mp*-CeO₂ has a well-defined mesoporous structure replicated from SBA-15 (Figure 1a,b). The *mp*-CeO₂ has an internal concave surface of CeO₂ pores of *mp*-CeO₂ different from the external surface of rod-CeO₂. This internal concave surface of the CeO₂ pore of *mp*-CeO₂ is also different from the external surface of a whole *mp*-CeO₂ particle.³⁹ The TEM images of Au@*mp*-CeO₂ and Pt@*mp*-CeO₂ show that the metal nanoclusters are encapsulated in the *mp*-CeO₂ (Figure 2). This is different from the metal nanoclusters impregnated on the external surface of a *mp*-CeO₂ particle. Our catalytic measurements showed that the internal concave surface of the CeO₂ pores of *mp*-CeO₂ could be important for the WGS and thus reduce the activation energy. Thus the different surface structure of the internal concave surface of CeO₂ pores of *mp*-CeO₂ could result in different catalytic behaviors.

To build a correlation between the measured difference in activation energies and the potential difference in surface chemistry of different CeO₂, we performed *in situ* studies of these catalysts by using AP-XPS in reduction with H₂ at 300 °C or under WGS reaction conditions: Au@*mp*-CeO₂, Pt@*mp*-CeO₂, Cu@*mp*-CeO₂, Pd@*mp*-CeO₂, Au/rod-CeO₂, Pt/rod-CeO₂, Cu/rod-CeO₂, and Pd/rod-CeO₂.

Figures 5–8 present the photoemission features of Au and Pt nanoclusters encapsulated in channels of *mp*-CeO₂ and supported on rod-CeO₂ catalysts during WGS. As described in the Methods section, XPS studies of M@*mp*-CeO₂ were performed on samples upon Ar sputtering. The same photoemission features of core levels of metals were observed upon sputterings of different amounts of time. It suggested that the metal nanoclusters were certainly encapsulated in CeO₂ pores of *mp*-CeO₂ (Figure S3c). Otherwise, there would not be photoemission features of core levels of metal

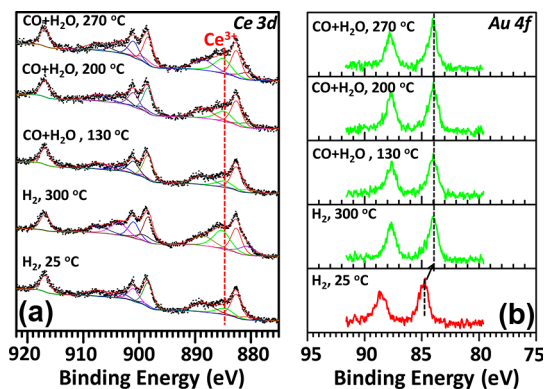


Figure 5. Photoemission features of (a) Ce 3d and (b) Au 4f of Au@*mp*-CeO₂ under reaction conditions studied with AP-XPS. $P_{\text{CO}} = 0.3$ Torr, $P_{\text{H}_2\text{O}} = 1.2$ Torr, $P_{\text{H}_2} = 1$ Torr. Reaction temperatures are given in the figure.

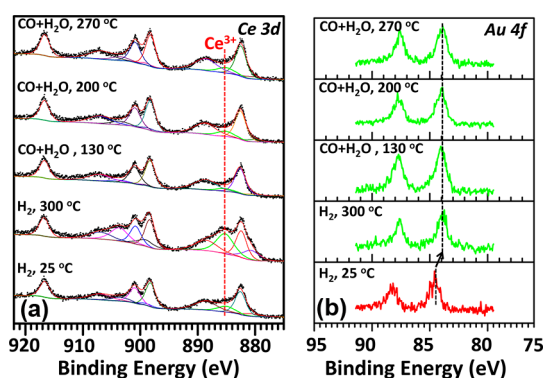


Figure 6. Photoemission features of (a) Ce 3d and (b) Au 4f of Au/rod-CeO₂ under reaction conditions studied with AP-XPS. $P_{\text{CO}} = 0.3$ Torr, $P_{\text{H}_2\text{O}} = 1.2$ Torr, $P_{\text{H}_2} = 1$ Torr. Reaction temperatures are given in the figure.

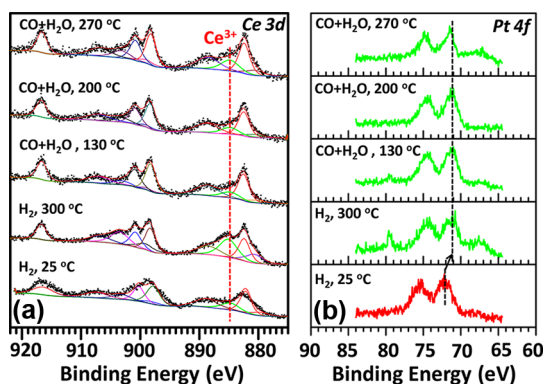


Figure 7. Photoemission features of (a) Ce 3d and (b) Pt 4f of Pt@*mp*-CeO₂ under reaction conditions studied with AP-XPS. $P_{\text{CO}} = 0.3$ Torr, $P_{\text{H}_2\text{O}} = 1.2$ Torr, $P_{\text{H}_2} = 1$ Torr. Reaction temperatures are given in the figure.

nanoclusters after removal of a certain shell region of M@*mp*-CeO₂ through sputtering (Figure S3).

Spectra of Ce 3d of the eight catalysts under reaction conditions are listed in Figures 5a, 7a, 8a, 9a, S7, S8, S9, and S10. Typically, a Ce 3d spectrum can be deconvoluted into photoemission features of Ce³⁺ and Ce⁴⁺.⁴⁰

Due to a final state effect, there are five sets of 3d spin–orbit split doublets. The peaks denoted by u''' , u'' , u , v''' , v'' , and v are attributable to Ce^{4+} , which are dominant in these catalysts (Figures 5–8 and S7–S10). Meanwhile, the u' , u^0 , v' , and v^0 peaks are assigned to Ce^{3+} . For all of the Ce 3d spectra, the same constraint was implemented during their deconvolutions. The peak positions and widths of these components in different spectra strictly remained consistent. For each spin–orbital splitting, the energy split remained constant for catalysts with different support (*mp*- CeO_2 or rod- CeO_2). The implementation of these identical parameters makes the comparison of calculated Ce^{3+} atomic fractions of different spectra collected under different reaction conditions eligible.

The evolution of oxygen vacancy density presented by Ce^{3+} fractions in Figure 9 suggests a different surface chemistry of internal concave walls of the CeO_2 pore of *mp*- CeO_2 of $M@mp\text{-CeO}_2$ in contrast to the external surface of rod- CeO_2 of $M/\text{rod-CeO}_2$ under the same reaction condition. For example, the fraction of Ce^{3+} in $M@mp\text{-CeO}_2$ under a WGS condition is typically 17–28% or even larger. However, it is about 4–12% in the case of $M/\text{rod-CeO}_2$ under the same WGS condition. A similar difference was exhibited between

$\text{Pd}@mp\text{-CeO}_2$ and $\text{Pd}/\text{rod-CeO}_2$ and between $\text{Cu}@mp\text{-CeO}_2$ and $\text{Cu}/\text{rod-CeO}_2$, as well.

The fraction of Ce^{3+} in $\text{Au}@mp\text{-CeO}_2$ and $\text{Pt}@mp\text{-CeO}_2$ is ~ 17 and $\sim 16\%$ before a WGS reaction (Figure 9). Upon a treatment of H_2 at 300 °C, density of oxygen vacancies (represented with the fraction of Ce^{3+} in the total of Ce^{3+} and Ce^{4+}) increases to $\sim 39\%$ (measured in the presence of H_2 at 300 °C). During WGS catalysis, however, the density of oxygen vacancies remains at 18–28% for $\text{Au}@mp\text{-CeO}_2$ and 16–27% for $\text{Pt}@mp\text{-CeO}_2$. It is suggested that H_2O dissociatively adsorbs on oxygen vacancies, decreasing density of oxygen vacancies created in the H_2 pretreatment. Under the WGS reaction condition, the Ce^{3+} atomic fraction in $\text{Au}/\text{rod-CeO}_2$ and $\text{Pt}/\text{rod-CeO}_2$ decreases to 4–12% only. Clearly, the density of oxygen vacancies in the surface region of rod- CeO_2 of $M/\text{rod-CeO}_2$ is lower than that of the surface region of internal concave walls of the CeO_2 pore of $M@mp\text{-CeO}_2$ during WGS catalysis. A similar difference was observed between $\text{Pd}@mp\text{-CeO}_2$ and $\text{Pd}/\text{rod-CeO}_2$ and between $\text{Cu}@mp\text{-CeO}_2$ and $\text{Cu}/\text{rod-CeO}_2$ (Figures S7–S10).

Typically, H_2O dissociates on oxygen vacancies of bifunctional WGS catalysts (metal nanoclusters supported on reducible oxides or clusters of reducible oxide supported on the metal surface).¹⁶ Higher adsorption energy of OH groups on oxygen vacancies means a stronger adsorption of OH groups during catalysis from the point of view of thermodynamics. Thus, a lower density of oxygen vacancies could be observed if there is a stronger adsorption of OH groups. The reason that OH groups are favorably adsorbed on oxygen vacancies of a rod- CeO_2 instead of a concave internal surface of CeO_2 pores of *mp*- CeO_2 and the adsorption energy of a OH group on an oxygen vacancy of a rod- CeO_2 is stronger is a larger repulsion between adjacent OH groups on the concave internal surface resulting from the limited space of the concave internal surface. This difference is schematically shown in Figure S12. The concave internal surface increases repulsion of the neighboring adsorbates. The increase of repulsion of OH groups on the internal concave

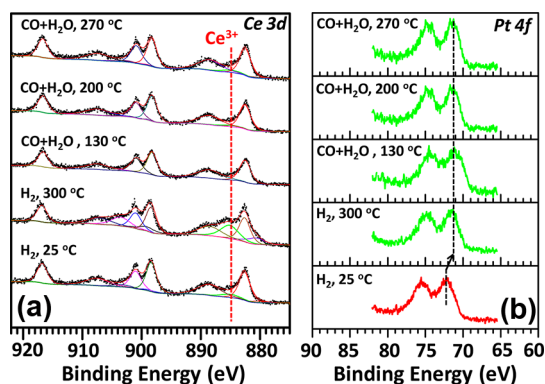


Figure 8. Photoemission features of (a) Ce 3d and (b) Pt 4f of $\text{Pt}/\text{rod-CeO}_2$ under reaction conditions studied with AP-XPS. $P_{\text{CO}} = 0.3$ Torr, $P_{\text{H}_2\text{O}} = 1.2$ Torr, $P_{\text{H}_2} = 1$ Torr. Reaction temperatures are given in the figure.

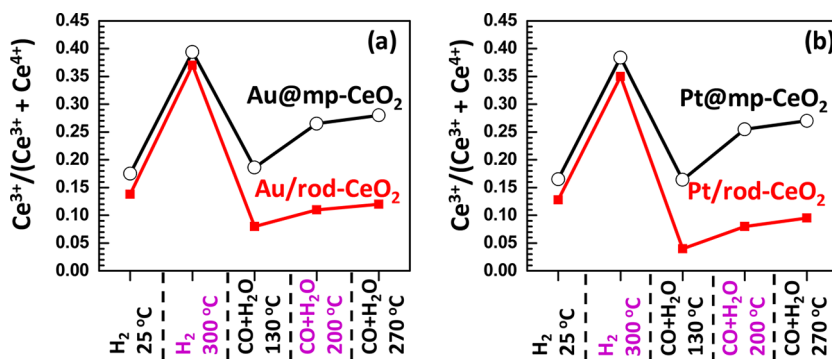


Figure 9. Evolution of atomic fractions of Ce^{3+} of $\text{Au}@mp\text{-CeO}_2$ and $\text{Au}/\text{rod-CeO}_2$ (a) and $\text{Pt}@mp\text{-CeO}_2$ and $\text{Pt}/\text{rod-CeO}_2$ during WGS.

surface is an opposite effect to the decrease of repulsion of adsorbates through formation of an open surface of nanoclusters *via* breaking a stepped surface Pt(557).^{26,27}

XPS studies show that both gold and platinum of metal nanoclusters *in mp-CeO₂* or *on rod-CeO₂* before reactions exist in an oxidized chemical state instead of a metal state (Figures 5–8). They were reduced to the metallic state upon the pretreatment in H₂ at 300 °C. Notably, AP-XPS studies showed the binding energies of Au 4f and Pt 4f remain in a metallic state under WGS conditions (Figures 5–8). As described in the Methods section, the surface charging is in the range of about 0.2 eV or less. In addition, the difference in surface charging between a catalyst in H₂ and one in CO + H₂O is extremely small. Thus, the downshift of binding energy of Au 4f (or Pt 4f) in the WGS condition in contrast to that before any reactions clearly indicated a change of oxidation state of gold (or platinum) in these catalysts during catalysis. Thus, the *in situ* studies using AP-XPS suggested the metallic state of Au and Pt encapsulated in *mp-CeO₂* or *on rod-CeO₂* during WGS catalysis. The metallic state of Pd and Cu was also suggested for Pd@*mp-CeO₂* (Figure S7), Pd/rod-CeO₂ (Figure S8), Cu@*mp-CeO₂* (Figure S9), and Cu/rod-CeO₂ (Figure S10) under WGS reaction conditions.

These parallel studies of M@*mp-CeO₂* and M/rod-CeO₂ catalysts show a higher density of oxygen vacancies on the surface of the internal concave surface of CeO₂ pores of *mp-CeO₂* during catalysis (Figure 9). In addition, AP-XPS studies suggested the metallic state

of Au, Pt, Pd, and Cu nanoparticles encapsulated in *mp-CeO₂* during catalysis. On the basis of these *in situ* studies, the difference in activation energy is related to the different surface chemistry of *mp-CeO₂* of M@*mp-CeO₂* and rod-CeO₂ of M/rod-CeO₂ under the same reaction condition. The high concentration of oxygen vacancies on the internal wall of *mp-CeO₂* suggests low adsorption energy of the OH group on the internal concave wall of *mp-CeO₂* during catalysis, which could contribute to the lower activation energy of WGS on M@*mp-CeO₂* compared to M/rod-CeO₂.

SUMMARY

Au, Pt, Pd, and Cu nanoclusters encapsulated in *mp-CeO₂* were synthesized through a template of SBA-15. WGS performed at the interface of Au, Pt, Pd, and Cu nanoclusters and the concave surface of the internal walls of CeO₂ pore of *mp-CeO₂* exhibits lower activation energies than metal nanoclusters supported on flat surfaces of rod-CeO₂. The surface chemistries of metal nanoclusters and CeO₂ during WGS were studied by using AP-XPS. Our AP-XPS studies identified the metallic state of Au, Pt, Pd, and Cu nanoclusters of these catalysts *during* WGS. The measurements of oxygen vacancies of *mp-CeO₂* of M@*mp-CeO₂* and rod-CeO₂ of M/rod-CeO₂ during WGS suggested a correlation between the surface chemistry of the concave, surface of the internal walls of the CeO₂ pore of *mp-CeO₂*, and low activation energy of M@*mp-CeO₂* in contrast to those of M/rod-CeO₂.

METHODS

Synthesis of M@*mp-CeO₂* involved three steps (synthesis of SBA-15, formation of *mp-CeO₂* using the SBA-15 template, and encapsulation of metal nanoclusters into channels of *mp-CeO₂*), as schematically shown in Figure 10. It is different from the deposition–precipitation method for formation of metal nanoparticles *on mp-CeO₂* particles³⁹ since the metal nanoclusters in our work are encapsulated *in the channel of mp-CeO₂*.

The *mp-CeO₂* was prepared by the hard-template method. SBA-15 was used as the template.⁴¹ The SBA-15 was synthesized by following the method reported in the literature.³ Typically, 10.2 g of Pluronic P123 (EO₂₀PO₇₀EO₂₀ (*M_{av}* = 5800) Aldrich) was dissolved into 329 mL of deionized water and 62 mL of 37% HCl solution. Then, 22.44 g of tetraethylorthosilicate (TEOS) (Aldrich) was added to the acid solution and was subsequently stirred at 35 °C for 20 h. Then the solution was crystallized at 80 °C for 10 h. The crystallization yielded white precipitates, which were collected by filtration. The white precipitates were further calcined at 550 °C for 3 h to remove organic surfactants. The as-obtained product is SBA-15 (Figure 10a), which is used as the hard template for preparation of *mp-CeO₂*.

In the synthesis of *mp-CeO₂*, 9.4 g of Ce(NO₃)₃·6H₂O was dissolved into 20 mL of ethanol (Aldrich). Then, 2 g of SBA-15 was added to the ethanol solution kept at 35 °C with mild stirring to evaporate the ethanol. After evaporation, the mixture was calcined at 300 °C for 3 h to transform the Ce(NO₃)₃ to CeO₂. Afterward, the impregnation process was repeated by using 3.3 g of Ce(NO₃)₃·6H₂O in 20 mL of ethanol. This product was calcined at 550 °C for 3 h to form *mp-CeO₂*/SBA (Figure 10b). The silica in the *mp-CeO₂*/SBA was removed by 1 M NaOH solution for three times to get *mp-CeO₂* (Figure 10c), which was dried at 110 °C overnight. Figure 10 schematically represents the

process of synthesizing *mp-CeO₂*.⁴¹ The crystallization of *mp-CeO₂* was confirmed with SEM and XRD (Figure 1).

For the synthesis of rod-CeO₂, a hydrothermal method was used.⁴² 0.868 g of Ce(NO₃)₃·6H₂O was dissolved in 5 mL of deionized water. Then 35 mL of 6 M NaOH was added to the solution. The as-obtained mixture was transferred and sealed into a Teflon-lined autoclave and kept at 100 °C for 24 h. The precipitate was collected by filtration and calcination at 400 °C for 4 h, and the final product was denoted as rod-CeO₂.

The metal nanoclusters were formed on the internal surface of the wall of channels of *mp-CeO₂* or on the external surface of rod-CeO₂ through impregnation. A certain amount of precursors of these metal nanoclusters was dissolved in ethanol, and 0.5 g of *mp-CeO₂* or rod-CeO₂ was added to the solution. HAuCl₄·3H₂O (Aldrich), H₂PtCl₆·6H₂O (Aldrich), PdCl₂ (Aldrich), and Cu(NO₃)₂·2.5H₂O (Aldrich) were precursors of impregnation for formation of Au, Pt, Pd, and Cu metal nanoclusters in CeO₂ pores of *mp-CeO₂* and on rod-CeO₂. The loading of each metal on CeO₂ is 2.5 wt %. Precursor is dissolved in ethanol. The ethanol was gradually evaporated at 35 °C, then the solids were calcined at ~450 °C for 3 h. Size of metal nanoclusters encapsulated in *mp-CeO₂* is similar to those on rod-CeO₂ (see Table 1).

Powder XRD patterns of catalysts were collected on a Bruker D8 Avance diffractometer using Cu K α radiation ($\lambda = 1.54056 \text{ \AA}$). Structure and morphology of catalysts synthesized above were characterized with SEM (FEI Magellan XHR 400) at 15 kV. Size, shape, and lattice fringe of the catalysts upon a pretreatment in H₂ were identified with Titan TEM (FEI Titan 80–300, 300 kV FEG TEM with point resolution 0.2 \AA).

The catalytic activity of the WGS was tested in a fixed-bed microflow reactor at atmospheric pressure. Fifty milligrams of

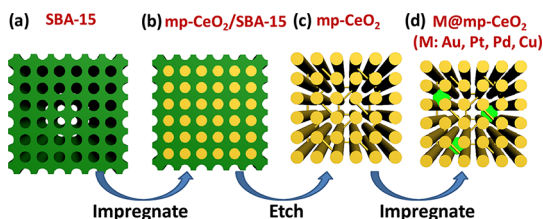


Figure 10. Process of the synthesis of metal nanoclusters encapsulated in pores of $mp\text{-CeO}_2$ ($M@mp\text{-CeO}_2$). The green balls in (d) represent metal nanoclusters encapsulated in $mp\text{-CeO}_2$.

catalyst was used for each test. Before catalytic tests, WGS catalysts were reduced *in situ* under a flow of 5% $\text{H}_2/95\%\text{Ar}$ (50 mL min^{-1}) at $300\text{ }^\circ\text{C}$ for 45 min. After the pretreatment, the catalysts were cooled to room temperature in 5% H_2/Ar (50 mL min^{-1}). Reactant composition was 5.6% CO and 22.44% H_2O with Ar balance. Water was brought into the reactor by a high-performance liquid chromatography pump. The total flow rate of the reactant gas is 50 mL min^{-1} . This catalytic condition kept the conversion lower than 5% for kinetic studies. The effluent gas was analyzed by a gas chromatograph (HP 5890 II). The CO conversions were calculated using following equations:

$$X\% = \frac{A_{t_0} - A_t}{A_{t_0}}$$

Here A is the peak area of CO from the GC; t_0 is the sampling time before the WGS reaction, and t is after the WGS reaction. Plotting of $\ln X$ as a function of $1000/T$ gave the slope, from which activation energies E_a were calculated.

In situ studies of metal nanoclusters encapsulated in $mp\text{-CeO}_2$ and supported on rod- CeO_2 were performed by using the ambient-pressure XPS using a benchtop X-ray source, monochromated Al $K\alpha$. This system includes three sections: ambient-pressure XPS, sample preparation chamber, and 20 bar reaction cell (not used for this work). Figure S1 shows photos of this system in the Tao group. Reactant gas can flow through the catalysts during XPS analysis. During catalysis, samples are heated conductively through an external heating source. This external heating excluded the possibility of any catalytic reactions that a filament may induce. The test of this whole AP-XPS system showed that it can examine a silver thin-film sample at a pressure up to 25–50 Torr. The resolution of Ag $3d_{5/2}$ in 25 Torr N_2 is 0.46 or 0.85 eV corresponding to a pass energy of 5 or 40 eV, respectively. The reasonable count rate of ~ 6000 at a pressure as high as 25–50 Torr N_2 suggests the capability of *in situ* studies at such a relatively high pressure (Figure S2). The acquisition time of Ag $3d$ (Figure S2) within a few minutes exceeded the possibility of reducing sample surfaces due to a long-time X-ray irradiation. Detailed information on the AP-XPS system and its functions can be found in Supporting Information and literature.⁴³

To avoid surface charging, we do not use a silicon wafer to load catalysts for AP-XPS studies. Here, a Au or Ag single crystal is used as a substrate for AP-XPS studies. A Au single crystal was used as a substrate of Pt@ $mp\text{-CeO}_2$, Pt/rod- CeO_2 , Cu@ $mp\text{-CeO}_2$, and Cu/rod- CeO_2 samples for AP-XPS studies. Due to the overlap of the photoemission features of Au $3d$ and Pd $3d$, a Ag single crystal was used as a substrate of Au@ $mp\text{-CeO}_2$, Au/rod- CeO_2 , Pd@ $mp\text{-CeO}_2$, and Pd/rod- CeO_2 . Catalysts were dispersed in ethanol. Samples of AP-XPS studies were prepared by drop-casting the ethanol solution containing the dispersed WGS nanoparticle catalyst onto the Au or Ag single-crystal surface. Natural evaporation of ethanol in ambient condition at $50\text{ }^\circ\text{C}$ was performed before loading a sample into the UHV chamber and then transferred to the AP-XPS reaction cell.

To minimize potential surface charging during data acquisition, coverage of nanocatalysts on the surface of Au or Ag is kept at a level lower than 40%. The charging is in the range of 0.2 eV, which is nearly within error of binding energy measurements of XPS using Al $K\alpha$.

It is well-known that the sampling depth of AP-XPS is very similar to vacuum XPS in the range of a few nanometers. To

exclude the potential contribution of metal nanoclusters potentially formed on the external surface of a $mp\text{-CeO}_2$ particle to the photoemission feature, surface layers (about 10 nm) were removed through Ar sputtering. The removal of a certain thickness can be confirmed with electron microscopy techniques. To confirm the removal of nanoclusters potentially formed on the surface of $mp\text{-CeO}_2$ particles, a longer sputtering was typically performed. There was no change of photoemission features and intensities of core levels from metal nanoclusters upon a longer sputtering. So, we exclude the possibility of contribution of information from nanoclusters potentially formed on the external surface of $mp\text{-CeO}_2$ particles. Then, the sputtered surfaces were used for XPS studies. Figure S3 schematically presents the method described above.

The AP-XPS studies of $M@mp\text{-CeO}_2$ using AP-XPS were performed at temperatures of 130, 200, and $270\text{ }^\circ\text{C}$ in a gaseous environment of 0.3 Torr CO and 1.2 Torr H_2O , or 1 Torr of H_2 . Argon ion sputtering was used to remove any potential metal nanoparticles on the external surface of $mp\text{-CeO}_2$ particles to make sure the photoemission feature of metal nanoclusters is solely contributed from the nanoclusters encapsulated in $mp\text{-CeO}_2$. In addition, TEM studies of the edge of $mp\text{-CeO}_2$ by tilting the edge of $M@mp\text{-CeO}_2$ during TEM examination did not find metal nanoparticles on the external surface of $M@mp\text{-CeO}_2$.

Deconvolution of all Ce 3d spectra was done with Casa always using the same set of parameters. The positions, widths, and shapes of 10 peaks for each Ce 3d spectrum were fixed so that the calculated atomic ratios of Ce^{3+} to a total of Ce^{3+} and Ce^{4+} are comparable. Each of these Ce 3d spectra was deconvoluted into 10 peaks: $\nu_{\text{O}}, \nu', \nu'', \nu''', \mu_{\text{O}}, \mu', \mu'', \mu'''$ from low binding energy to high binding energy. The six peaks corresponding to three pairs of spin-orbital doublets ($\nu, \nu', \nu'', \mu, \mu', \mu''$ and ν''', μ''', μ'''') are contributed from Ce^{4+} , resulting from different Ce 4f occupancies in the final state.⁴⁴ The $\nu_{\text{O}}, \nu', \mu_{\text{O}}$ and μ' are assigned to Ce^{3+} .³⁴ The positions of the six peaks of Ce^{4+} and four peaks of Ce^{3+} referred to XPS studies of CeO_2 and Ce_2O_3 samples reported in the literature⁴⁰ and from our recent studies of pure oxides. The measured nominal binding energies of μ'''' of Ce^{4+} in our XPS studies are very close to that reported in literature⁴⁰ within the error of measurements, which confirmed the minimized surface charging under reaction conditions in our experiments.

Conflict of Interest: The authors declare no competing financial interest.

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Supporting Information Available: Characterization of catalysts during catalysis using in-house ambient-pressure XPS, XRD, and TEM of rod- CeO_2 and other catalysts, TEM images of $M@mp\text{-CeO}_2$ and $M/\text{rod-}\text{CeO}_2$, their photoemission features studied using AP-XPS, and structural model illustrating the difference between external surface of rod- CeO_2 and concave internal surface of $mp\text{-CeO}_2$ are available. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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